

# Growth, morphology and structure of CsRP<sub>4</sub>O<sub>12</sub> crystals

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We have grown CsRP<sub>4</sub>O<sub>12</sub> (R<sup>3+</sup> is a rare-earth ion) crystals from 85% phosphoric acid solutions. Morphological, optical and X-ray studies have been carried out on all the crystals. CsRP<sub>4</sub>O<sub>12</sub> crystallizes in four different structural modifications. The variation in growth temperature causes variation in their morphology and crystal structure.

## 1. Introduction

Six types of rare-earth phosphates have been reported so far: orthophosphates [R<sup>3+</sup>PO<sub>4</sub>, M<sub>3</sub>R<sup>3+</sup>(PO<sub>4</sub>)<sub>3</sub> and M<sub>3</sub>R<sup>3+</sup>(PO<sub>4</sub>)<sub>2</sub>], pyrophosphates [R<sub>4</sub><sup>3+</sup>(P<sub>2</sub>O<sub>7</sub>)<sub>3</sub>], tripolyphosphates [R<sup>3+</sup>H<sub>2</sub>P<sub>3</sub>O<sub>10</sub>], metaphosphates [R<sup>3+</sup>(PO<sub>3</sub>)<sub>3</sub>], ultraphosphates [R<sup>3+</sup>P<sub>5</sub>O<sub>14</sub>] and tetraphosphates [M<sup>1</sup>R<sup>3+</sup>(PO<sub>3</sub>)<sub>4</sub>], where M is Li, Na, K, Rb, Cs or NH<sub>4</sub>; R<sup>3+</sup> is a rare-earth ion. Orthophosphates, metaphosphates and ultraphosphates containing the rare-earth ions have been prepared by many workers, for example, Beucher [1], Hong [2], Bonder and Tananaev [3] and Kizilyalli [4], but only a few rare-earth tetraphosphates are known, although many of them serve as important source materials in optoelectronics. We have made an attempt, therefore, to grow Cs tetraphosphates containing all the rare-earth ions by slow heating of aqueous solutions containing Cs<sub>2</sub>CO<sub>3</sub> and various rare-earth oxides in 85% phosphoric acid. The present paper reports the growth of CsRP<sub>4</sub>O<sub>12</sub> crystals with special reference to their morphological, optical and X-ray characterization.

## 2. Crystal growth

We have studied phase formation in the system Cs<sub>2</sub>O-R<sub>2</sub>O<sub>3</sub>-P<sub>2</sub>O<sub>5</sub>-H<sub>2</sub>O within the temperature range 350 to 670°C and at a constant partial pressure of water (~1 atm) in order to grow CsR tetraphosphate monocrystals. A mixture of 85% phosphoric acid and Cs<sub>2</sub>CO<sub>3</sub> was placed in a vitreous carbon glass crucible and it was held at room temperature till evolution of CO<sub>2</sub> was complete. Later, corresponding R<sub>2</sub>O<sub>3</sub> was added to this

mixture and the crucible was kept inside the furnace. The rare-earth oxide dissolves completely in 85% orthophosphoric acid at nearly 250°C to form a homogeneous solution. It is well known that CsR tetraphosphates show a negative temperature coefficient of solubility in phosphoric acid. Therefore, this homogeneous solution was held at 250 to 300°C for 1 to 2 days until the desired equilibrium between vapour and solution had been established. Then the temperature of the reactor (furnace) was slowly increased to a predetermined temperature at a rate of 3.5 to 4°C per hour and held for 7 to 8 days. A more detailed account of the growth is given by Litvin *et al.* [5]. The CsRP<sub>4</sub>O<sub>12</sub> crystals were obtained using different concentrations of Cs<sub>2</sub>O = 27 to 45 wt%; P<sub>2</sub>O<sub>5</sub> = 52 to 70 wt%.

The crystals resulted from homogeneous nucleation and they range in size from 1 to 3 mm. We found that the variation in the ratio of the nutrient components did not influence the size of the crystals. The results of our experiments are shown in a composition diagram showing fields of crystallization for CsRP<sub>4</sub>O<sub>12</sub> crystals in the system Cs<sub>2</sub>O-R<sub>2</sub>O<sub>3</sub>-P<sub>2</sub>O<sub>5</sub>-H<sub>2</sub>O (Fig. 1). The changes in composition of Cs<sub>2</sub>O and P<sub>2</sub>O<sub>5</sub> determine boundaries of crystallization for various phases. Good quality crystals can be obtained by selecting suitable concentrations for Cs<sub>2</sub>O and P<sub>2</sub>O<sub>5</sub> within the boundaries indicated. The crossed region in the composition diagram represents fields of simultaneous crystallization of neighbouring phases. The study of optical, morphological and X-ray characterization revealed four different modifications of CsRP<sub>4</sub>O<sub>12</sub> crystals.

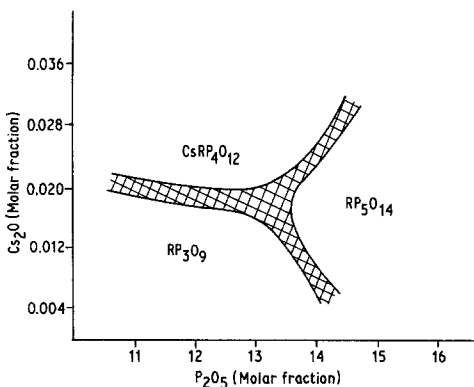


Figure 1 Diagram showing starting compositions of solutions.

### 3. Crystal morphology

The morphology of  $\text{CsRP}_4\text{O}_{12}$  crystals is given in Table I. On the basis of the morphology of these crystals, we conclude that the tetrahedral faces (Fig. 2a) characteristic of crystals grown at lower temperature ( $\sim 350$  to  $500^\circ\text{C}$ ) with rare-earth ions of higher ionic radii are replaced by tabular, prismatic and clinohedral faces (Fig. 2b and c) in crystals containing rare-earth ions of lower ionic radii. The crystals showing prismatic and clinohedral forms are extended longitudinally parallel to the  $c$ -axis. The crystals grown at higher temperatures ( $> 520^\circ\text{C}$ ) are characterized by prism and pyramidal forms (crystals with rare-earth ions from La to Er, Fig. 2d and e) and clinohedral forms (crystals with rare-earth ions from Tm to Lu). With a further increase in growth temperature

( $> 680^\circ\text{C}$ ), the crystals become smaller and more equi-dimensional (Fig. 2f) due to the higher ionic mobility, low viscosity and higher concentration of the solute ions. Plate-like crystals are not observed in the tetraphosphates, contrary to what is observed in the ultraphosphates.

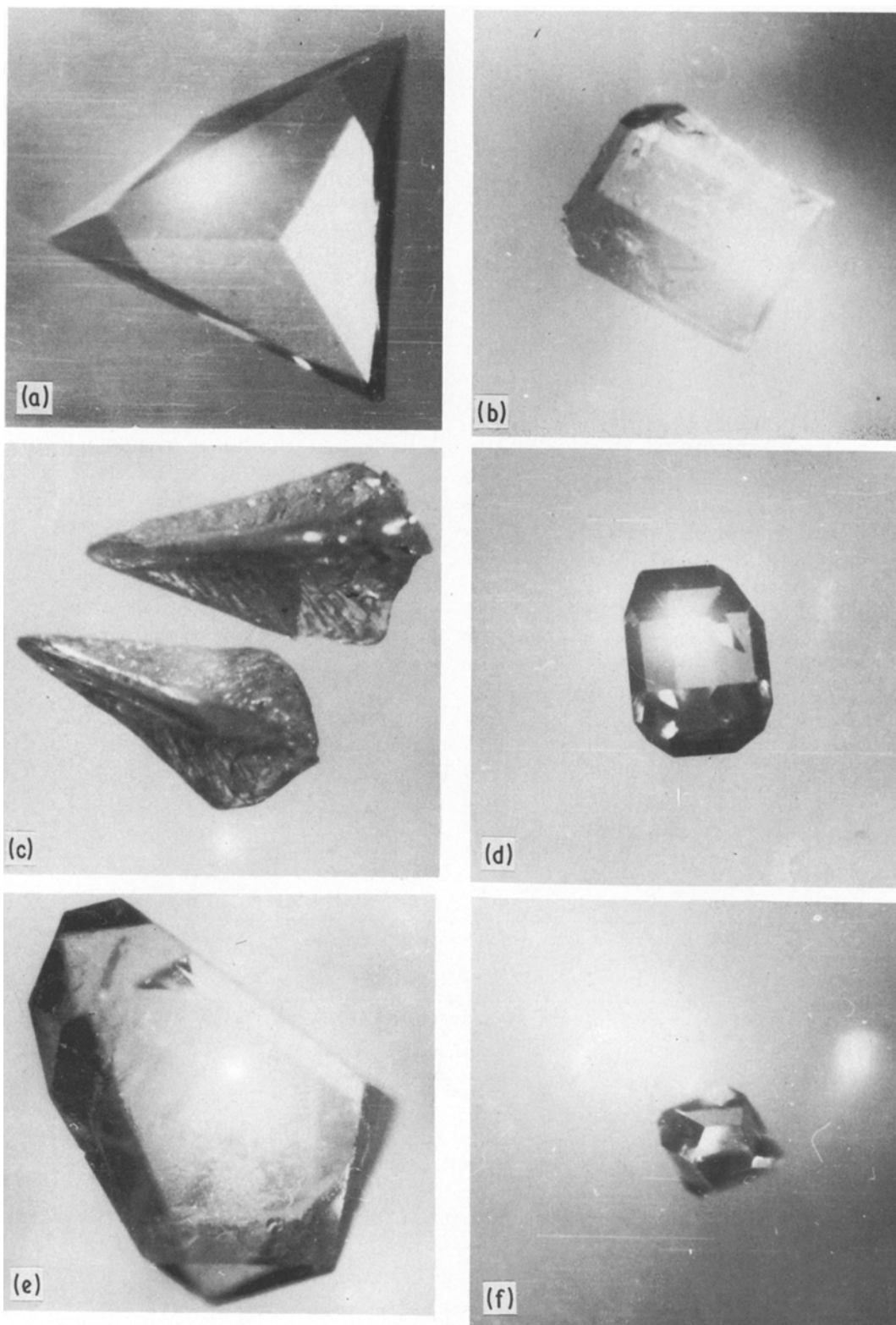
The crystals belonging to the second structure type (crystals containing rare-earth ions from La to Nd) resemble an ideal crystal of monazite (Fig. 2d). These crystals were obtained at  $580^\circ\text{C}$  and they show a diamondiferous lustre. The faces are very well developed with highly polished surfaces. We have obtained these crystals using higher concentrations of  $\text{R}_2\text{O}_3$  and at higher temperatures of growth ranging from  $580$  to  $670^\circ\text{C}$ ; the quality of the crystals was exceptional. This does not agree with the results of Miyazawa *et al.* [6], who observed that higher temperatures of growth and higher concentrations of  $\text{R}_2\text{O}_3$  caused too rapid crystallization, resulting in poor quality and small size of  $\text{KNdP}_4\text{O}_{12}$  crystals. We observed that the variation in concentration of the nutrient components within the crystallization field of Fig. 1 has no influence on the size and optical quality of the crystals.

### 4. X-ray study

An X-ray diffractometer URS-50 IM with  $\text{CuK}\alpha$  radiation was used in the study of  $\text{CsRP}_4\text{O}_{12}$  crystals. Fig. 3 shows the dependence of the four different modifications, upon the ionic radii of the rare-earth ions and the growth temperatures. The

TABLE I Morphological characterization of  $\text{CsRP}_4\text{O}_{12}$  crystals

Rare-earth ions	Growth temperature ( $^\circ\text{C}$ )	Structure type	Morphology	Common faces in the crystals
La–Eu	350	I	Trigonal tetrahedral	(1 1 1), (0 1 1), (1 0 1)
Sm–Er	350	II	Tabular, prismatic, clinohedral and tetrahedral	(1 0 0), (1 1 1), (1 1 0), (0 1 0), (0 0 1), (1 0 1), (0 1 1), (1 $\bar{3}$ 1)
La–Nd	580	II	Tabular	(1 0 0), (1 1 0), (0 1 0), (1 0 1), (1 1 1), ( $\bar{1}$ 1 1)
Sm–Er	580	II	Prismatic, tabular and clinohedral	(1 1 0), (0 1 0), (0 0 1), (1 $\bar{3}$ 1), (1 1 1)
Tm–Lu	580	II	Clinohedral and tabular	(1 1 0), (0 1 1), ( $\bar{1}$ 1 0), (0 1 0), (1 $\bar{3}$ 1), (1 1 1)
La–Nd	720	II	Small equi-dimensional	(1 $\bar{1}$ 1), ( $\bar{1}$ 1 1), (0 1 0), (0 1 1)
Tm–Lu	350	III	Clinohedral and tabular	(1 1 0), (0 1 1), (0 1 0), ( $\bar{1}$ 1 0), (1 $\bar{3}$ 1), (1 1 1)
La–Nd	580	IV	Tabular, prismatic (poor development of the faces)	–



*Figure 2* Representative photographs of  $\text{CsRP}_4\text{O}_{12}$  crystals: (a)  $\text{CsNd}(\text{PO}_3)_4$  crystals from first structure type obtained at  $350^\circ\text{C}$ ; (b)  $\text{CsTb}(\text{PO}_3)_4$  crystals from second type obtained at  $350^\circ\text{C}$ ; (c)  $\text{CsLu}(\text{PO}_3)_4$  crystals from third structure type obtained at  $350^\circ\text{C}$ ; (d)  $\alpha\text{-CsNd}(\text{PO}_3)_4$  crystals from second structure type obtained at  $580^\circ\text{C}$ ; (e)  $\text{CsSm}(\text{PO}_3)_4$  crystals from second structure type obtained at  $580^\circ\text{C}$ ; and (f)  $\beta\text{-CsNd}(\text{PO}_3)_4$  crystals from fourth structure type obtained at  $720^\circ\text{C}$ .

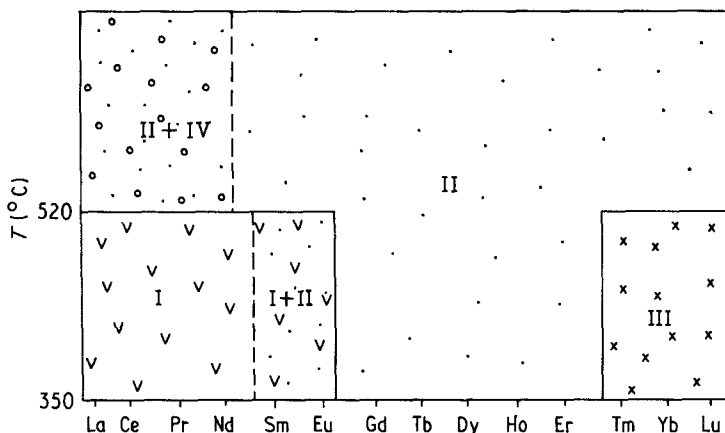


Figure 3 Different structural types of  $\text{CsRP}_4\text{O}_{12}$ , illustrating the relationship between the ionic radii of the rare-earth ions and the growth temperature: (I) Cubic, optically isotropic crystals; (II) Monoclinic, optically negative crystals; (III and IV) Monoclinic, optically positive crystals.

cell parameters for representative crystals from each group and their optical constants are given in Table II. The compounds with the rare-earth ions La to Eu form the first structure type, and the compounds with Sm and Eu ions occur in two modifications. Palkina *et al.* [7] reported the cubic modification of  $\text{CsRP}_4\text{O}_{12}$ , where  $R = \text{Nd}$  and  $\text{Pr}$ . We have determined the absolute cell parameters for all the compounds of this structure type. The values of the cell parameters were plotted against the ionic radii of the rare-earth ions (Fig. 4) which shows a linear dependence between the two. However,  $\text{CsCeP}_4\text{O}_{12}$  proved to be an exception. The values of ionic radii for the rare-earth ions are taken from Ahrens [8].

For the second and third structure types of  $\text{CsRP}_4\text{O}_{12}$  crystals, we determined the cell parameters of  $\text{CsGd}(\text{PO}_3)_4$  and  $\text{CsLu}(\text{PO}_3)_4$ , respectively. The cell parameters of the latter have not been reported in the literature, therefore, a detailed structural analysis of this modification is being carried out by us.

For the fourth structure type, the cell parameters of  $\beta\text{-CsNd}(\text{PO}_3)_4$  are given in Table II. With a further increase in growth temperature, this structure type continues to crystallize. Thus, the X-ray study revealed that  $\text{CsRP}_4\text{O}_{12}$  crystallizes in four different modifications. Those containing rare-earth ions La to Nd form three different modifications and those containing the remainder

TABLE II Cell parameters and optical constants for  $\text{CsRP}_4\text{O}_{12}$  crystals

Structure type	Crystal composition	Cell parameters		Optical constants	
		Structure	Space group	Refractive indices	$2V(^{\circ})$
I	$\text{CsNd}(\text{PO}_3)_4$	Cubic $a = 15.232 \text{ nm}$	$I4_132^*$	1.533	—
II	$\text{CsGd}(\text{PO}_3)_4$	Monoclinic $a = 10.449 (3) \text{ nm}$ $b = 13.057 (4) \text{ nm}$ $c = 8.930 (3) \text{ nm}$ $\gamma = 56.13^{\circ}$	$P2_1/b$	$N_g = 1.632$ $N_m = 1.627$ $N_p = 1.621$	— 50
III	$\text{CsLu}(\text{PO}_3)_4$	Monoclinic $a = 9.955 (3) \text{ nm}$ $b = 13.961 (3) \text{ nm}$ $c = 20.023 (5) \text{ nm}$ $\gamma = 52.39^{\circ}$	$B2/b$	$N_g = 1.573$ $N_m = 1.562$ $N_p = 1.556$	+ 85
IV	$\beta\text{-CsNd}(\text{PO}_3)_4$	Monoclinic $a = 7.145 \text{ nm}$ $b = 8.809 \text{ nm}$ $c = 9.146 \text{ nm}$ $\gamma = 99.65^{\circ}$	$P2_1$	$N_g = 1.636$ $N_m = 1.630$ $N_p = 1.617$	+ 68

\*See Palkina *et al.* [3].

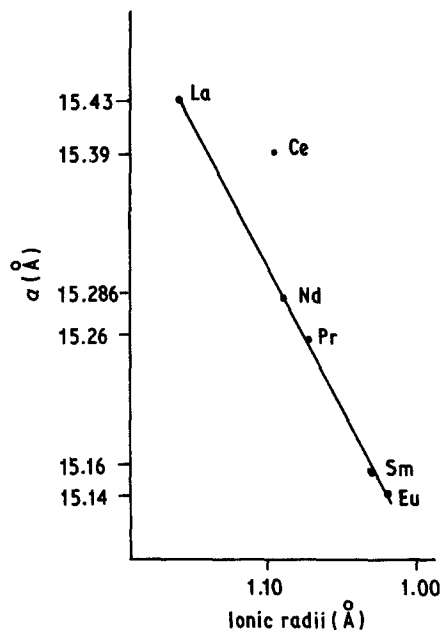


Figure 4 Linear dependence of the cell parameters and ionic radii of the rare-earth ions (for first structure type of the crystals).

of the rare-earth ions crystallize in two different modifications (Fig. 3).

## 5. Conclusions

(a) Morphological, X-ray and optical studies reveal four different structural groups among the  $\text{CsRP}_4\text{O}_{12}$  crystals.

(b) The variation in growth temperature and ionic radii of the rare-earth ions changes the morphology and crystal structure.

(c) Three structural modifications of  $\text{CsRP}_4\text{O}_{12}$  occur with the rare-earth ions La to Nd depending upon the growth temperature, and two additional structural modifications occur with the rest of the rare-earth ions.

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